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# Selective Pattern Transfer of Nano-Scale Features Generated by FE-SPL in 10 nm Thick Resist Layers

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**Abstract:** High performance single nanometer lithography is an enabling technology for beyond CMOS devices. In this terms a novel mask- and development-less patterning scheme by using electric field, current controlled Scanning Probe Lithography (FE-SPL) in order to pattern structures on different samples was developed. This work aims to manufacture nanostructures into different resist by using FE-SPL, whereas plasma etching at cryogenic temperatures is applied for an efficient pattern transfer into the bottom Si substrate. The challenge for future quantum devices, generated by SPL and cryogenic etching, is finding a resist that is at most 10 nm in thickness and has a plasma durability high enough for pattern transfer into silicon. As a first step towards future quantum devices the silicon-to-resist selectivity of calixarene, AZ Barli, poly (3-hexylthiophen-2, 5-diyl) and polymethylmethacrylat for the anisotropic cryogenic dry etching process was estimated. A silicon-to-resist selectivity of about 4:1 for each of these resists was found. With these results, nano-scale, highly parallel double line features in silicon for future double patterning were generated.

**Keywords:** Field Emission Scanning Probe Lithography, Cryogenic Plasma Etching, Single-Electron Devices

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## 1. Introduction

Scaling down of device sizes has been the fundamental strategy for improving the performance and efficiency of nanoelectromechanical devices and systems (NEMS). Field-Emission Scanning Probe Lithography (FE-SPL) permits a precise and cost effective fabrication route towards the enabling of new devices as, e.g., single electron transistors [1]. Thereby, a Fowler-Nordheim-type electron emission of low energy electrons from a sharp scanning tip is caused by an intense electric field [2]. In contrast to EBL, in which high energy electrons are generating secondary electrons which induce the lithographic reaction, in FE-SPL electrons in an energy range (typically < 50eV) just needed for chemical

interaction with the resist are generated. The tone of patterning can be switched by variation of the exposure dose of electrons [2]. Thus, for obtaining positive tone lithography a wet development step is no more required [2]. Furthermore, due to small tip-sample distances and low electron energies FE-SPL can take place in ambient conditions without the requirement for vacuum or special gaseous environments. An atomic force microscopy (AFM) imaging directly after exposure can be done using the same scanning probe, termed 'active cantilever'. This allows the inspection of the generated features in a closed loop imaging scheme. The closed loop FE-SPL offers thus single-nanometer manufacturing and has the potential to replace conventional lithography techniques in terms of resolution, direct

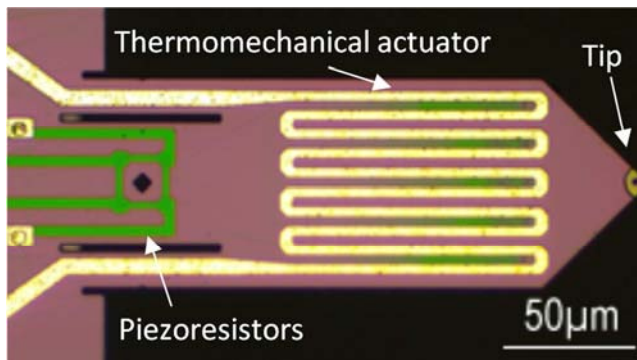
patterning, and tone switching without post processing [3, 4].

Moreover, a reliable dry etching recipe operating at cryogenic temperatures for pattern transfer into silicon using a relatively thin resist layer of calixarene, (usually with thicknesses of 10 to 50 nm) and preserving the lithographic resolution was developed. Thereby, the combination of cryogenic temperatures with the reactive ion etching using SF<sub>6</sub> with a certain amount of oxygen enables a highly anisotropic and pattern preserving etching. The process is established for aspect ratios and linewidths below 10 nm and for high aspect ratio (HAR) in the 50-100 nm regime respectively.

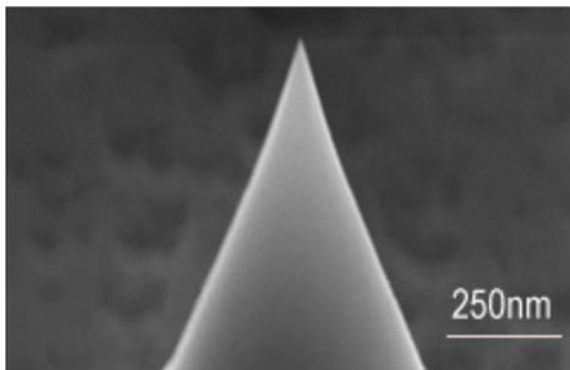
In this work results from the optimized lithography and etching protocols as well as tests of the plasma durability and etching stability of different resists typically used for lithography purposes will be presented.

## 2. Field-Emission Scanning Probe Lithography

For pattern generation of nano-scale features into ultrathin resists, field-emission scanning probe lithography (FE-SPL) is used. An applied bias voltage between tip and sample generates an electric field. Due to the enhancement of the field caused by the ultrasharp tip [2], a constant Fowler-Nordheim electron emission can occur. These electrons are used to generate features in the resist.



(a)



(b)

**Figure 1.** (a) SEM image of the thermomechanically actuated piezoresistive cantilever. (b) A sub-10nm write and imaging resolution by SPL/AFM require cantilever tips with radius of curvature of < 15 nm [5].

The utilization of self-actuated, self-sensing active cantilevers with a resonance frequency of about 100 kHz, enables not only to structure the resist with FE-SPL but also to inspect the written features by AFM. The oscillation of these cantilevers is induced thermomechanically, whereas the deflection is measured by piezoresistors. The thermomechanical actuator, as indicated in Figure 1 (a) by yellow color, actuates the cantilever on the basis of the bimorph effect. It consists of a metallic micro heater, which excites the cantilever at its resonance, and simultaneously deflects the cantilever at lower frequencies off-resonance (1 Hz to 20 kHz). In this respect, the cantilever bends proportionally to the square of the input DC-current to the micro-heater. The deflection read-out of the cantilever is realized by piezoresistive sensors integrated on the base of cantilever beam (colored green in Figure 1 (a)) [5- 9]. Four piezoresistors are arranged in an integrated Wheatstone bridge configuration to reduce the influence of temperature variations and noise. At the end of the cantilever beam a sharp conductive tip with typically < 15 nm tip radius of curvature is integrated (Figure 1 (b)). The shape of the tip has considerable influence on the electric field and the trajectories of the electrons [10], as well as on the emission probability [2]. A more detailed information about the active cantilever, their fabrication process as well as their performance is provided in [9].

The use of self-sensing and self-actuated scanning probes enables the implementation of two independent feedback loops in the setup: The first is the feedback loop for the AFM, to image patterns in high resolution and perform surface inspection. Here the height of the scanning probe is regulated in dependence of the amplitude of oscillation. The second is a current feedback for FE-SPL, where the height of the scanning probe is changed to keep the current setpoint. With the combination of these 2 loops, the surface can be inspected and structured with the same scanning probe.

The SPL tool operates in constant-current Fowler-Nordheim regime, thus pattern generation into the resist is done by a fine beam of low energy electrons (a few eV < E < 50 eV) and results in either development-less positive tone or negative tone behavior, dependent on the type of resist and the electron dose. The principle can be seen in Figure 2. The linewidth of the generated patterns can be regulated by the electron dose. To adjust the favored electron dose, the FE-SPL tool is capable of setting both, current and writing speed. The resulting line dose can be calculated as

$$LD = \frac{I}{v},$$

where  $LD$  is the line dose,  $I$  is the current between sample and tip and  $v$  is the writing speed. The tool normally operates in a range for currents and writing speeds of 3 pA – 100 pA and 0.1 μm/s – 10 μm/s, respectively.

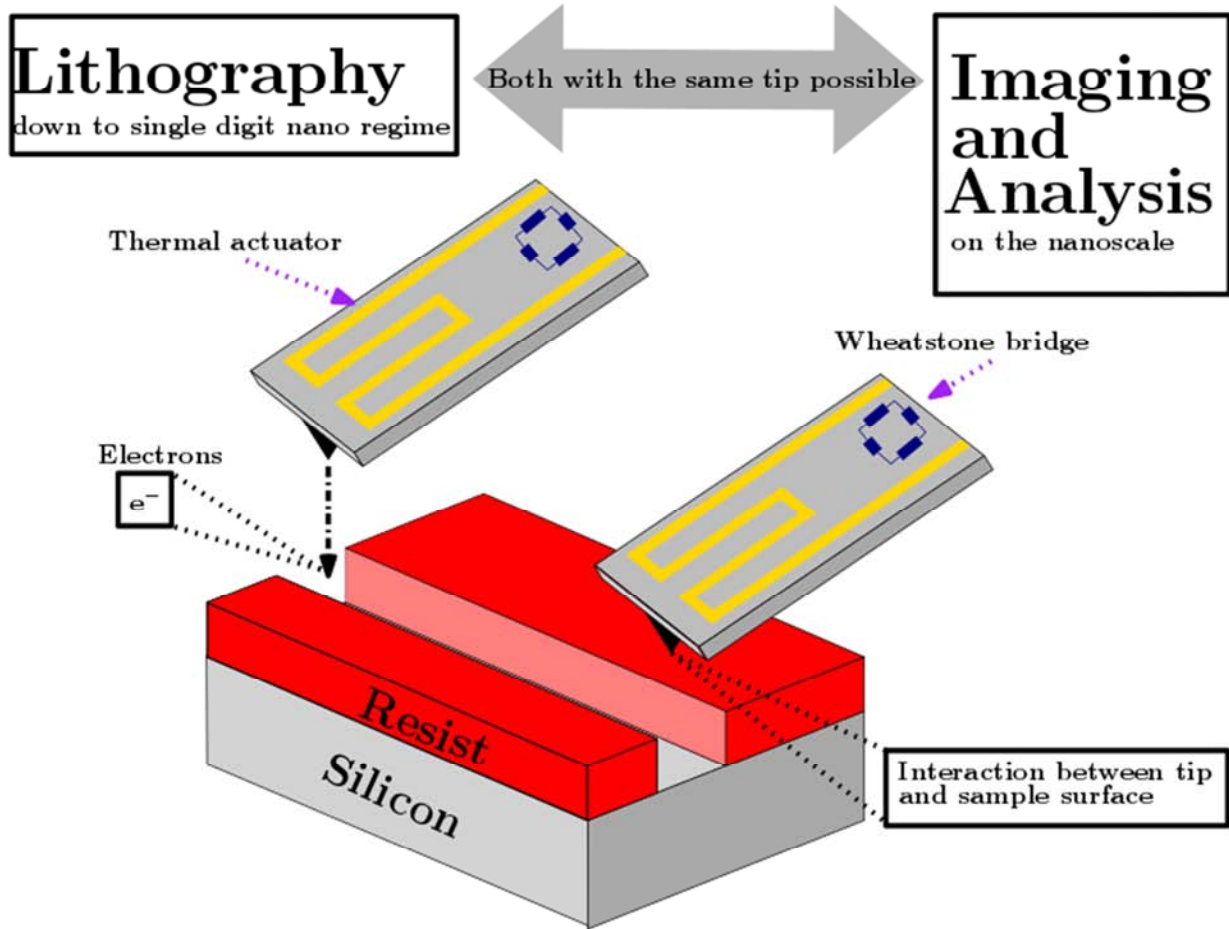


Figure 2. Concept of closed-loop for imaging and patterning. Figure according to [4].

Beside the electron dose, the distance between tip and sample is important for a stable lithographic process. The tip-to-sample distance can be regulated by the applied voltage (at a constant current setpoint), whereby a higher voltage causes a higher distance and vice versa. The bias voltage can be regulated in the range of 5 V to 100 V. The optimal bias voltage value thereby depends on the resist, resist thickness, substrate and tip.

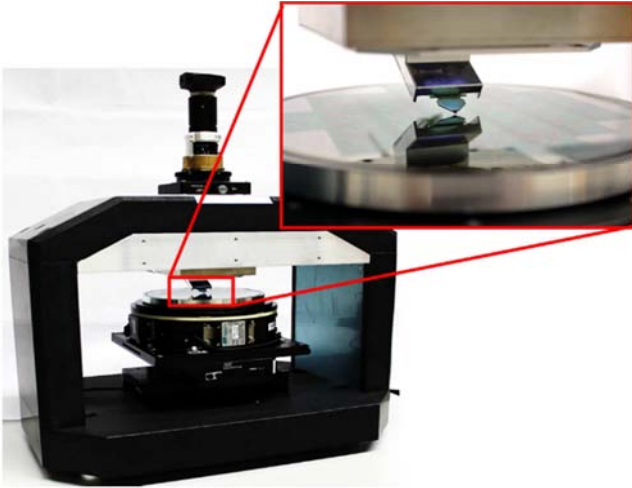
It should be mentioned that the environmental conditions should be stable during the lithography process. The optimal temperature and humidity is in the range of  $22^{\circ}\text{C} \pm 2^{\circ}\text{C}$ ,  $35\% \pm 10\%$ , respectively. Large fluctuation of the environmental conditions can cause a change of tone or even a tip crash. The general setup of the FE-SPL tool is shown in Figure 3. The system is based on a granite block and a damping system to minimize thermal and mechanical drift. The top scanner for the regulation of the scanning probe height is mounted on a cross-beam suspension for higher mechanical stability. Two different scanners can be applied; one providing a  $10 \times 10 \mu\text{m}^2$  scan area and another with  $200 \times 200 \mu\text{m}^2$  scan area. The sample is located on a coarse positioning bottom stage to enable an increased patterning area of up to  $100 \times 100 \text{mm}^2$  size. A step-and-repeat function enables an active pattern area up to 4 inch wafers [1]. The step-and-repeat function is mostly used to generate same

structures repeatedly on the sample. By optical navigation through a camera mounted on the top of the system, structures can be found easily.

The aim of the FE-SPL is

- (a) scaling down the nanomanufacturing to the single nanometer,
- (b) development of nano-lithographic methods with respect to cost efficiency and reliability,
- (c) to realize ultra-low power electronics, single electron transistors, quantum devices and individual atom manipulation and
- (d) to create new technology for beyond CMOS devices with help of cost-efficiencies, to implement mix-and-match technique.

In contrast to EBL, the proximity effect is strongly reduced due to the low energy of the electrons used in FE-SPL, which enables to write features with half-pitches below 10nm and line widths below 5 nm. The capability of patterning single nanometer structures is opening a new horizon for future nanoelectronics. Theoretically, scanning probes are able to image and pattern structures down to the atomic scale [11, 12]. In this respect, quantum devices, i.e. single-electron devices working at room-temperature, could already be manufactured [1].



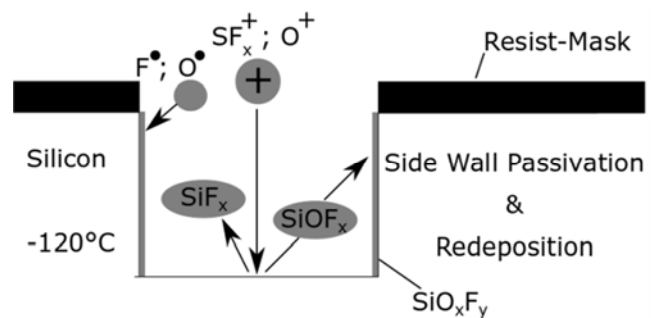
**Figure 3.** Cross-beam setup combining top scanner and bottom coarse positioning stages enabling a step-and-repeat-based SPL scheme.

### 3. Cryogenic Etching

For a pattern transfer in the nanoscale range, a cryogenic anisotropic etching procedure is used. This process has been first introduced by Tachi *et al.* [13] and many studies [14-16] revealed enhancements of the cryogenic etching process compared to standard reactive ion etching (RIE). Basically, the idea of cryogenic etching is to combine the effect of ionized feed gases with low substrate temperatures [17]. The low temperature increases the sticking probabilities of the reactants and enables the formation of a polymeric layer on the silicon surface [18]. In this work,  $\text{SF}_6/\text{O}_2$  gases at cryogenic temperatures are used. These feed gases are commonly used for deep anisotropic etching of silicon. In this particular process, both oxygen and the low temperature are necessary to produce the passivation layer at the feature sidewalls. By lowering the temperature below  $-100^\circ\text{C}$ , recombination of Si, O, and F becomes energetically favorable and  $\text{SiO}_x\text{F}_y$  molecules are created. Furthermore, the attack of sidewalls by radicals and the evaporation rate of halogen compounds such as  $\text{SiO}_x\text{F}_y$  products are thermally activated by an Arrhenius law for activation energy, hence these rates are reduced when the substrate temperature is lowered [18]. This results in a polymeric  $\text{SiO}_x\text{F}_y$  passivation layer on the silicon surface at cryogenic temperatures. This layer protects the silicon from the spontaneous etching by the fluorine. Due to the condensation of  $\text{Si}_2\text{F}_6$  products on the silicon surface, the spontaneous etching of Si by fluorine is blocked at temperatures lower than  $-196^\circ\text{C}$  [19]. Thus the practically usable temperature is limited. For the cryogenic etching process, the typical temperature range is between  $-100^\circ\text{C}$  to  $-140^\circ\text{C}$  [17].

To this point, the passivation layer acts as an etch stop, because the layer hinders further etching in both vertical and horizontal direction. Thus, additionally to the chemical etching, a sputtering effect is applied by a potential difference between plasma bulk and sample surface. This triggers an acceleration of ions straight towards the sample [20]. These accelerated ions will have energies of a few tens

of electron-volt, depending on the actual process conditions. The vertically accelerated ions have a glancing angle of incidence to the feature sidewalls, resulting in low or no sputtering of the  $\text{SiO}_x\text{F}_y$  layer. On the contrary, at the bottom of the etched feature the sputtering yield is high enough to effectively remove any polymer already deposited. Additionally, the physical sputtering due to the ion bombardment also prevents the growth of an inhibitor film. Thus the fluorine can react spontaneously on the bottom of the feature, forming volatile  $\text{SiF}_4$  molecules. This attack of silicon by fluorine is an exothermal chemical reaction. Finally, the etching product then desorbs from the surface. As a result of the above described processes, the etching proceeds only in vertical direction (ion bombardment direction) and an anisotropic profile can be achieved. [13, 18, 20-23]. Figure 4 depicts the principle of cryogenic etching.

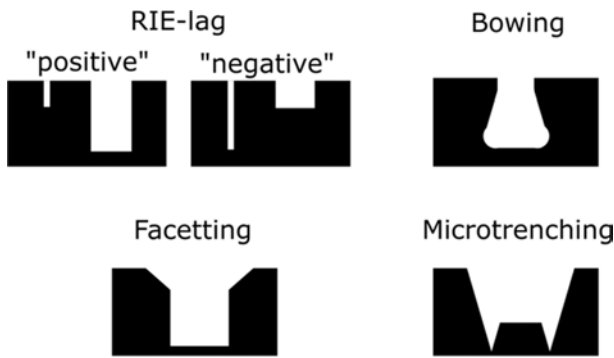


**Figure 4.** Demonstration of basic principle of ions and radicals involved in cryoetching. Figure according to [22].

In 2005 Mellhaoui *et al.* published a cryogenic etching test for different substrate temperatures, with or without oxygen and ion bombardment [24]. They concluded that the injection of  $\text{O}_2$  to the  $\text{SF}_6$  gas results in the formation of perfect passivation layers on the feature sidewalls, whilst prevailing cryogenic temperature. When the temperature is increased to room temperature after etching, the  $\text{SiO}_x\text{F}_y$  passivation layer becomes volatile and leads to smooth silicon sidewalls. Thus, cryogenic etching allows precise pattern transfer for sub 20 nm features with high aspect ratio. It should be noted, that it has been demonstrated, that the passivation can also be created by the injection of  $\text{SiF}_4$  and  $\text{O}_2$  [25]. Chopping-mode etches for high aspect ratios using  $\text{SF}_6$  and  $\text{O}_2$  have also been demonstrated [26]. Since cryogenic etching is basically a reactive ion etching (RIE) process, it suffers from the same high aspect ratio effects. These effects can only be observed for features with aspect ratios  $>3:1$ . The most significant of these aspect ratio dependent effects (ARDE) are RIE-lag, bowing (barrelling), faceting and microtrenching (see Figure 5). These effects are caused by several mechanisms:

- Ion and kinetic neutral flux
- Electron shading effect
- Transport and Depletion of chemical etch and inhibit reactants

For further details and informations about ARDE see [18, 27].



**Figure 5.** The most significant aspect ratio dependent effects. Figure according to [18].

## 4. FE-SPL Resists

Both, lithography and the subsequent etching are affected by the resist. In this work, the influence of the pattern transfer by cryogenic etching on the FE-SPL defined pattern, more specific, the selectivity, was in the spotlight. Thus the selectivity and plasma durability of different standard FE-SPL resist will be compared.

For the creation of nanoscale features, a molecular resist is favored compared to conventional resists. Typically, calixarene resist is used for FE-SPL. Calixarenes belong to the group of metacyclophanes, are cup-shaped and the molecule size is about 1 nm. Due to its small particle size, low molecular weight, precise defined spatial extent, monodisperse nature and high uniformity, these resists are capable of a higher resolution and lower line edge roughness for the SPL process. Furthermore, calixarene resists offer the possibility for tone switching by electron dose changes. Due to its thermal and chemical stability [28], calixarene can be coated on rough and non-flat surfaces. It also provides high plasma etch durability, which is an important factor for the production of nanodevices [29].

**Table 1.** Mixing ratio and solvent for each resist taken in this work. It should be noted, that AZ Barli is weighed in ml (1 ml AZ Barli in 13 ml ELL).<sup>1</sup>

Resist type	Amount of resist [mg]	Solvent	Amount of solvent [ml]	Layer thickness [nm]
Calixarene	70	MCB	10	20
Calixarene	35.49	MCB	10	10
AZ Barli	1	ELL	0	200
AZ Barli	1	ELL	13	10
P3HT	4	Toluol	2	10
PMMA	36.91	MIBK	10	10

Ohnishi et al. handles calixarene resists as “prospective material for nanofabrication” [30]. The following resists were compared: AZ® Barli®, Poly (3-hexylthiophen-2, 5-diyl) (short: P3HT) and Polymethylmethacrylat (short: PMMA).

PMMA is a standard resist for EBL. Details about the lithographic resolution for polymeric resists, in particular PMMA, can be found in [31]. AZ Barli is an antireflective coating for photolithography but is used as resist for FE-SPL

as well, since it offers a small surface roughness. Due to the lithography mechanism it is necessary that either the sample or the resist is conductive. Unfortunately, all of the resist types introduced to this point are insulating. A promising loophole for future FE-SPL with non-conductive chips could be P3HT, which is a p-type semiconductor and is mainly used as part of the photoactive layer in organic solar cells or as semiconducting layer in organic field-effect transistors [32].

## 5. Methodology

To optimize the etching protocol and determine the etching durability of different resists, different steps were conducted. Each resist is diluted in a solvent. The corresponding solvent and mixing ratio for each resist can be obtained from Table 1.

First, a resist layer was spin-coated onto n-doped silicon chips. The parameters for spin-coating and annealing can be found in Table 2. As second step, the resist layer was structured either by FE-SPL or scratching. Next, the patterned samples are cryogenically etched. Finally, the residual resist is removed by O<sub>2</sub>-plasma.

In this work, cryogenic etching was carried out with a PlasmaPro 100 Cobra ICP cryoetching tool from Oxford Instruments. In order to obtain a passivation layer, an oxygen gas flow of 2 sccm was used. Process pressure was 5 mTorr. A DC bias voltage of 70 V and RF at 11 Watt were kept constant. The wafer is fixed to the temperature-adjusted lower electrode. The substrate temperature was -120°C.

To estimate the selectivity of different resists calixarene, AZ® Barli®, P3HT and PMMA were spun on silicon chips and annealed them (see Table 2). In result, the samples had different resists, but same thicknesses of about 10 nm upon the respective silicon samples. Lines were scratched into the resist by a scalpel and measured by AFM to obtain the resist thickness. Afterwards, each sample was etched with our cryogenic etching process for 1 minute. Then, the residual resist was removed and the final structures were analyzed by AFM and the silicon-to-resist selectivities were estimated.

To determine if there are differences in the pattern transfer process for structures obtained by scratching or FE-SPL, an additional calixarene samples with a resist layer thickness of 21 nm was prepared. These samples were patterned by FE-SPL and then imaged by AFM. For FE-SPL an additional etching step was added to remove SiO<sub>2</sub>, which can be formed on the top layer of the Si during FE-SPL due to humidity. Next, the samples were cryogenically etched for 40 s, followed by resist removal and AFM measurements to obtain the final structure depth.

With the estimation of the silicon-to-resist selectivities, patterns and devices can be lithographed and etched. In this work the possibility to write perfectly parallel, self-aligned double line features via FE-SPL and cryogenic etching into silicon will be presented. Due to the gaussian beam, resist in the center of the electron beam is exposed with a higher exposure dose than on the outer beam areas. Combined with a resist capable to generate both, negative and positive tone, self-aligned double line features can be produced. The basic

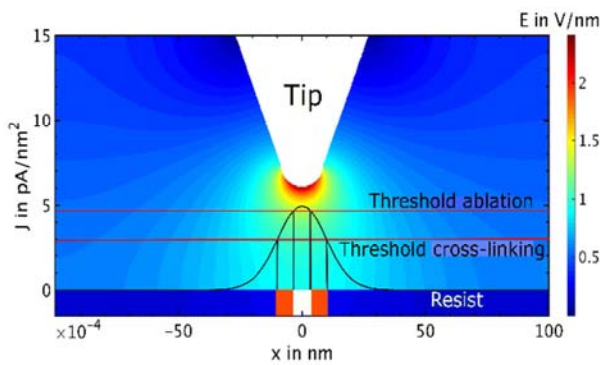
<sup>1</sup> MCB: Mono Chloro Benzene; ELL: Ethyl-L-Lactate; MIBK: Methyl isobutyl ketone

principle can be seen in Figure 6. Resist areas exposed with an exposure dose larger than the ablation threshold (middle of the electron beam) are directly ablated (positive tone). Resist areas exposed with an exposure dose between the ablation and the cross-linking threshold become cross-linked (negative tone). If

the resist is wet developed, only the cross-linked regions remain on the substrate, yielding thus self-aligned features. Their distance is thereby controlled by the ablation threshold in relation to the current density of the electron beam.

**Table 2.** Values for spin-coating and pre-bake of different resists.

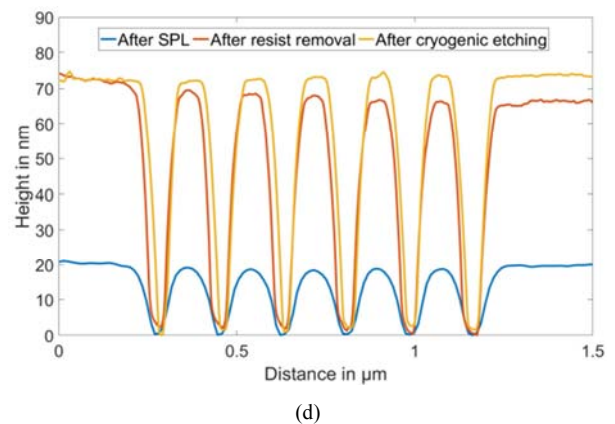
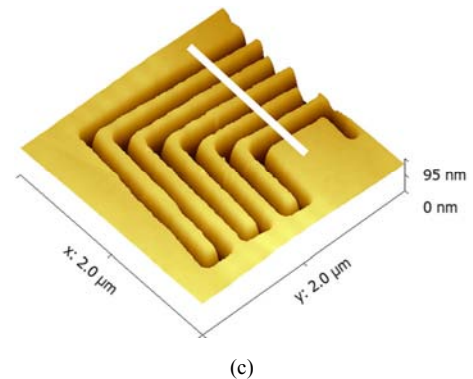
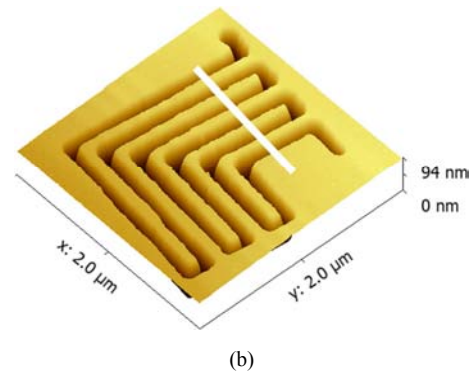
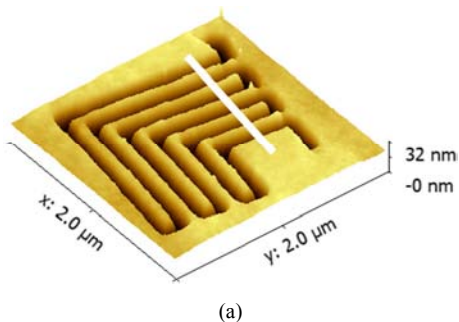
Resist type	Spin-coating				Annealing	
	1 <sup>st</sup> step		2 <sup>nd</sup> step		Annealing temperature °C	duration min
	Rotation speed rpm	duration s	Rotation speed rpm	duration s		
Calixarene	1200	3	2000	30	130	5
PMMA	1200	3	2000	30	130	5
AZ Barli	3000	30			200	1
P3HT	3200	45			125	5



**Figure 6.** Generation of double line features. Due to the gaussian distribution of the electrons, resist in the center of the electron beam is exposed with a higher exposure dose than the outer beam areas, resulting in direct ablation for an exposure dose higher than the “ablation threshold” (represented by the resist colored in white) and cross-linking between “ablation threshold” and “cross-linking threshold” (indicated by resist colored in red). For exposure doses below the “cross-linking threshold” the resist remains unaffected. After wet development only the red part of the resist remains on top of the substrate.

To demonstrate the process of preparing self-aligned double-line features, lines in a meander pattern shape were written into 9 nm calixarene resist with an SPL exposure dose of 200 nC/cm. Next, the resist was wet developed, followed by the cryogenic etching process (etching time was 25 s), as mentioned in the previous section, where the remaining resist was used as etching mask, and, afterwards, residual resist was removed by an oxygen plasma. Final structures were analyzed by AFM and SEM.

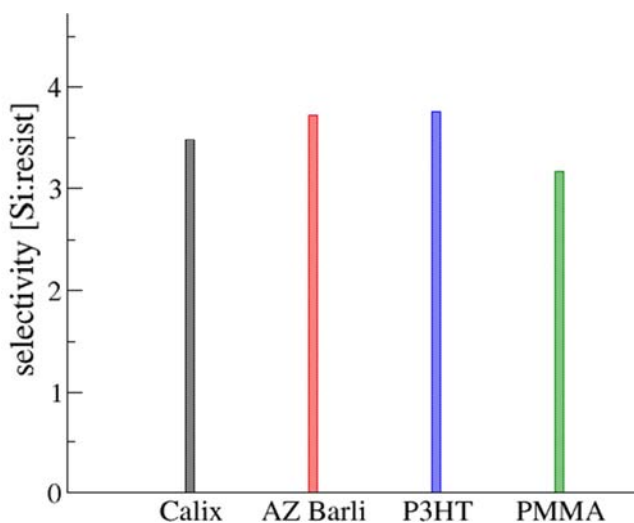
## 6. Results



**Figure 7.** AFM pictures of structures, after scanning probe lithography (a), after cryogenic etching (b) and after resist removal (c). The depth of all structures was obtained at the section indicated by the white line in the AFM picture. The AFM profile for the section can be seen in picture (d).

To obtain high-resolution FE-SPL patterns and fabricate future sub-10 nm structures a resist thickness of at most 10 nm will be required. Therefore, the plasma durability (silicon-to-resist selectivity) of different types of resists (calixarene, AZ Barli, P3HT and PMMA) is compared. The selectivities of the four resists were extracted from the AFM images taken after patterning, after etching and after resist removal. Examples of these AFM images in the different process steps [after FE-SPL (a), after cryogenic etching (b) and after resist removal (c)] are shown in Figure 7 for 21 nm thick calixarene resist patterned with FE-SPL.

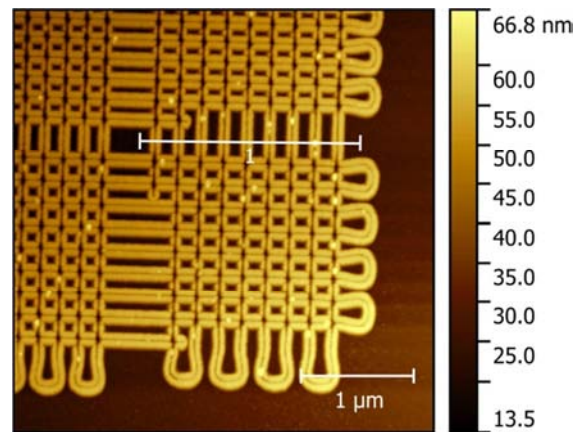
Figure 7 d) shows the AFM profile along the contour line in Figure 7 a)-c). The initial thickness of the resist is about 21 nm (obtainable from the contour “after lithography”), whilst after etching the depth has increased to approximately 71 nm. After removing the residual resist, the depth is around 68 nm. Thus approximately 3 nm of the resist layer were still on the silicon after the etching process. As result the silicon-to-calixarene selectivity can be estimated as  $\sim 3.5:1$ . Simultaneously it is possible to estimate the average etching rate of both, silicon and calixarene, for the cryogenic etching process used in this work, by dividing the etching depth by the etching time. The etching rates are around 1.7 nm/s and 0.5 nm/s, respectively. As can be seen from the AFM images {Figure 7 a)-c)} the shape of the pattern, line edge roughness and line width are preserved during the pattern transfer. It should be mentioned, that the measured line width is not the real line width, due to tip convolution in the AFM measurement. The resulting selectivities for the four different resists are shown in Figure 8. It can be seen, that the selectivities of the different resists are quite comparable. Thus their usability for fabrication of nanoscale features in silicon is only dependent on the FE-SPL resolution. While PMMA has a quite poor resolution, the other three resists (calixarene, AZ Barli and PHS) could all be used to prepare 10 nm half pitch structures in standard FE-SPL resolution tests.



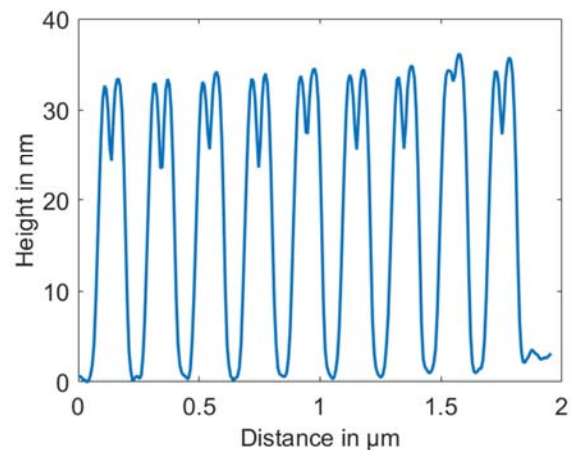
**Figure 8.** Silicon-to-resist selectivity of calixarene, AZ Barli, P3HT and PMMA. All selectivities were around 4:1.

To obtain the influence of the patterning technique on the etching results, two types of samples with calixarene resist were prepared by either scratching or FE-SPL, as described above. The selectivities of the samples with the different patterning techniques are comparable. Thus, it can be deduced that the patterning method is not influencing the etching result.

With the rough estimation of the silicon-to-resist selectivities, patterns and devices can be designed. To decrease the minimal pitch between two lines and obtain perfectly aligned, parallel lines, which is necessary for nanoelectronic devices, the possibility to create double-line features as described in the methodology section is explored.



(a)



(b)

**Figure 9.** (a): AFM picture of intersected double line features written via SPL and etched at cryogenic temperatures into silicon. (b): Contour of the measured double line features was obtained at the section indicated by the white line in the AFM picture. The height of the features is about 30 nm after etching. The measured total width is 130 nm, whereas the distance between the two parallel lines is about 30 nm.

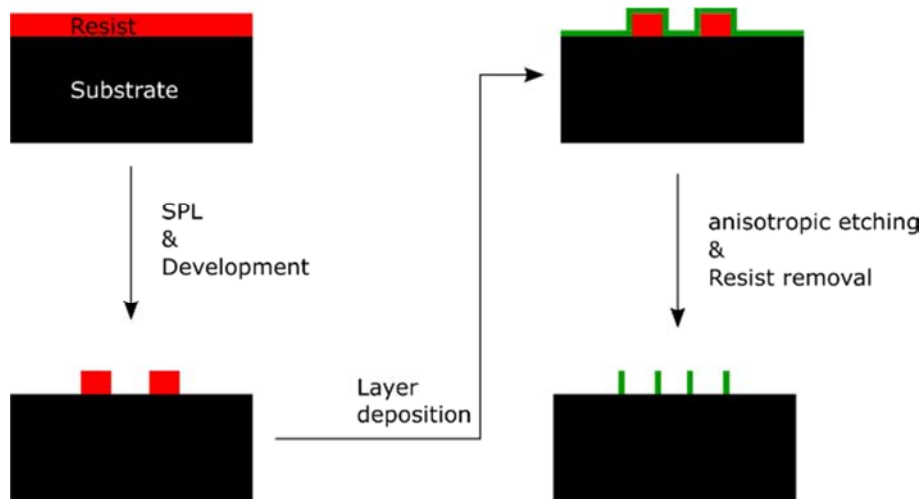
Therefore, single positive tone lines, as shown in Figure 9a, were written by FE-SPL (line dose was 200 nC/cm), then the resist was wet developed and finally cryogenically etched (etching time was 25 s). As can be seen from the AFM profile in Figure 9b self-aligned, perfectly parallel double line features are generated. The height of these

structures after etching is about 30 nm. The total width of these double lines is in the range of 130 nm, whereas the distance between the parallel lines is about 30 nm (etched trenches), resulting in a linewidth of a single line of about 50 nm. Line width and distance can be tuned by changing the FE-SPL exposure dose. The measured depth of such a trench is in the range of 5 nm (see Figure 9b), resulting in a discrepancy between total feature height (30 nm) and the measured trench depth (5 nm).

As can be seen from the SEM image of the generated double line features (Figure 10), the trench seems to be etched to the bottom. It can be concluded, that the measured trench depth of 5 nm is also just a measurement error due to the tip convolution in the AFM measurement.

As mentioned above, AFM pictures suffer from convolution of the real structure with the tip shape (tip radius around 10 nm). The result is an expanded measured structure. Furthermore, due to the tip size, it is also not able to fully penetrate into the etched trench between the two parallel lines. In this work no deconvolution was conducted, but there are different strategies to obtain the real feature after AFM [33- 35].

The double line features generated in this work could be used for pattern doubling, to further enhance the resolution capability of the lithography method. The use of the as produced double lines as spacer for “spacer defined double patterning” is another method to generate special nanostructures [36]. In Figure 11 the mechanism can be seen.

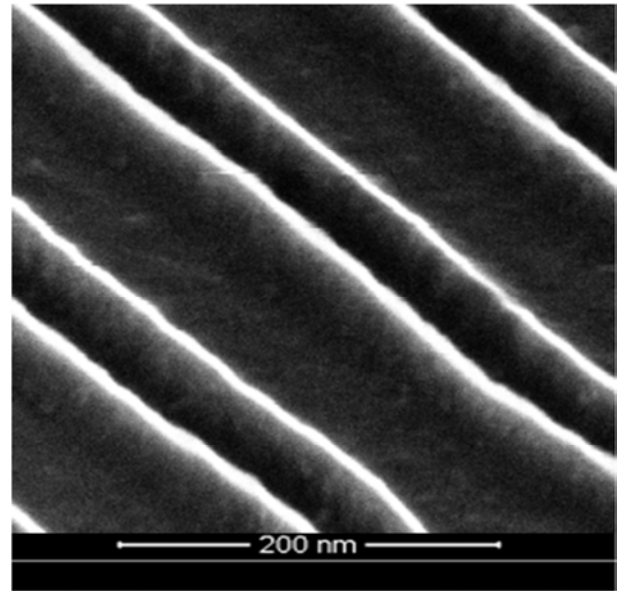


**Figure 11.** Procedure to use double line features for spacer defined double patterning.

## 7. Conclusion

Here, a method to estimate the silicon-to-resist selectivity (plasma durability) of different types of resist was presented. As a result, the molecular glass resist calixarene has a selectivity of about 4:1 to silicon. The plasma durability of calixarene, AZ Barli, P3HT and PMMA is comparable. Since calixarene, P3HT and AZ Barli offer similar FE-SPL resolutions (10 nm half pitch), the best resist for generation of a certain type of nanostructure can be chosen according to

After Scanning Probe Lithography and wet development, nanoscale double line features are generated. Subsequently, a second layer can be deposited on top of the features. After an anisotropic etch step, followed by resist removal, dense, highly parallel features in the nano scale can be produced on top of the substrate. Afterwards these features can be used as base for future production steps.



**Figure 10.** SEM picture of the etched double line features. A fully etched trench between the two parallel lines can be seen.

their individual properties as conductivity, tone switching and stiffness. Future high resolution lithography by FE-SPL and pattern transfer by cryogenic etching should be based on these resists. With an applied cryogenic etching step, field-emission scanning probe lithography offers a route for single nano-digit manufacturing and eventually could replace some cost-intensive lithography techniques in terms of resolution, patterning with low electron energy, direct patterning, tone switching, ambient conditions and costs.

In this work the generation of self-aligned, perfectly



parallel double line features in silicon by FE-SPL, wet development and cryogenic etching was demonstrated. These structures could be used for spacer defined double patterning in the future. Recently, single electron transistors were patterned on SOI-Wafer chips by FE-SPL, followed by a pattern transfer using cryogenic etching. The capability of these single electron transistors to work at room temperature was demonstrated [1]. Furthermore, we were able to generate templates with FE-SPL. These templates can be used for the production of high fidelity structures by nanoimprint lithography, enabling a high-throughput, high-resolution fabrication chain for future nanoelectronic devices like quantum dots and single electron devices [37].

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